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# Low toxicity antisolvent synthesis of composition-tunable luminescent allinorganic perovskite nanocrystals



CERAMICS

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#### ABSTRACT

All-inorganic perovskite nanocrystals have held great promise as incredibly effective luminescent materials and also have been synthesized efficiently by room temperature approaches. Toluene, a high toxicity solvent, is often used as a valid antisolvent in ligand-assisted reprecipitation strategy, which badly restricts the commercial application of this convenient method. Here, aiming to address the toluene toxicity issue, the low toxic tetraethyl orthosilicate (TEOS) is employed as a new alternative antisolvent to achieve high uniform and composition dependent luminescent all-inorganic perovskite  $Cs_4PbX_6$  (X = I, Br, Cl) combined with CsPbX<sub>3</sub> (X = I, Br, Cl) nanocrystals with tunable emission large-span wavelengths (626–437 nm) and remarkably narrow full width at half-maximum (FWHM) monochromaticity (down to 19 nm). Meanwhile, the ratio of CsBr to PbBr<sub>2</sub> is proved to be a critical factor to control the crystalline phase of the resulting perovskite nanocrystals. Additionally, one monochromatic light-emitting diode (LED) lamp with brightly pure green emission is fabricated based on bromine-based perovskite nanocrystals. This newly developed low toxicity antisolvent synthesis (LTaS) is suitable candidate for commercial production in an environmentally-friendly way, and the as-obtained perovskite materials with superior optical merits will be brought to the forefront of lighting and displays.

# 1. Introduction

Perovskites as leading candidates for optoelectronic devices exhibit intriguing semiconductor properties by virtue of long charge carrier diffusion lengths, high carrier mobility and long carrier life [1-6]. Depending on their impressive optoelectronic properties, perovskites have attracted considerable attention and also have been explored for potential applications in solar cells [7,8], light emitting diodes (LED) [9,10], photodetectors [11,12] and lasers [13,14]. Particularly organicinorganic hybrid perovskite solar cells have recorded efficiency higher than 21% recently [15], however, the inherent instability of these hybrid perovskites hinders the scaled production for practical applications [16-19]. Comparatively, Cs-based all-inorganic perovskites demonstrate superior physicochemical performance, especially including low trap state density, higher stability and versatile photoluminescence properties [20–22]. Protesescu et al. first reported the  $CsPbX_3$  (X = I, Br, Cl) nanocrystals via the hot-injection method, showing around 90% photoluminescence quantum yield (PLQY) as well as narrow emission peaks and wide color gamut since 2015 [23]. Subsequently, Li et al.

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introduced a novel supersaturated recrystallization synthesis operated at room temperature, the obtained  $CsPbX_3$  (X = I, Br, Cl) quantum dots are comparable with that synthesized by hot-injection method [24]. Taking advantage of ligand density control of quantum dots, Li et al. balanced surface passivation and carrier injection in  $CsPbX_3$  LED with a highest external quantum efficiency (EQE) of 6.27% [25]. Hence, emerging all-inorganic perovskite as a new emitter material will fill the impending demand on photoelectric devices, suggesting a luminous future for all-inorganic perovskite LED should be predictable.

Up to now, in order to satisfy the requirements of low cost and large scale production, facile room temperature method is widely employed for the preparation of all inorganic perovskites, which don't need relatively high reaction temperature, inert atmosphere and complex equipment. Sun et al. reported a ligand-mediated reprecipitation strategy accompanied with antisolvent, such as toluene and hexane, and the choice of organic acid and organic amine is critical for systematically manipulating the shape of  $CsPbX_3$  colloidal nanocrystals [26]. Recently, Chen et al. demonstrated hexagonal  $Cs_4PbBr_6$  synthesized by an inhomogeneous interface reaction at room temperature,

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which exhibited bright green emission with a large binding energy of 222 meV [27]. Analogously, tetragonal CsPb<sub>2</sub>Br<sub>5</sub> also has been successfully prepared by an alkyl-thiol ligand-assistant method at an ambient condition, and toluene is necessary for the formation of CsPb<sub>2</sub>Br<sub>5</sub> nanocrystals, which reflected high resistance to high temperature and high humidity environments [28]. For deeper understanding on intrinsic properties of perovskites with different dimensionalities, zerodimensional Cs<sub>4</sub>PbBr<sub>6</sub>, two-dimensional CsPb<sub>2</sub>Br<sub>5</sub> and three-dimensional CsPbBr<sub>3</sub> nanocrystals were achieved by a reverse microemulsion method at room temperature, and the phase transformation is triggered by cesium to lead ratio [29]. Generally, dimensionality refers to arrangement of lead halide octahedra in the perovskite crystal, the connectivity of lead halide octahedra dictates the optical behavior of perovskite materials, Pb<sup>2+</sup> and X<sup>-</sup> ions are related to the density of states in conduction and valance band respectively [30]. Although three perovskite-related phases can be obtained by simple room temperature method, most of the previous studies are limited to toluene and hexane as antisolvent, little other environmentally friendly antisolvent has been explored so far.

Herein, we developed a novel low toxicity antisolvent synthesis (LTaS) based on tetraethyl orthosilicate (TEOS) to obtain all-inorganic perovskite naocrystals at room temperature. Due to the low polarity of TEOS, the as-obtained all-inorganic perovskite nanocrystals presented excellently tunable large-span emission wavelengths (626–437 nm) and remarkable monochromaticity with narrow full width at half-maximum (FWHM) (low to 19 nm). Based on it, one typical monochromatic light-emitting diode (LED) lamp with brightly pure green emission was fabricated by employing UV LED chip (centered at 365 nm), evidently suggesting that the prepared perovskite nanocrystals with superior optical merits can be promisingly applied for lighting and also explored displays, and that this LTaS method should dramatically promote the commercial production of all-inorganic perovskite nanocrystals in an environmentally way.

# 2. Experiment section

### 2.1. Materials and chemicals

Lead (II) iodide (PbI<sub>2</sub>, 99.9%, aladdin), lead (II) bromide (PbBr<sub>2</sub>, 99.9%, aladdin), lead (II) chloride (PbCl<sub>2</sub>, 99.9%, aladdin), cesium (I) iodide (CsI, 99.9%, aladdin), cesium (I) bromide (CsBr, 99.9%, aladdin), cesium (I) chloride (CsCl, 99.9%, aladdin), oleylamine (OAm, 80–90%, aladdin), dimethylformamide (DMF, 99.8%, kelong), tetra-ethyl orthosilicate (TEOS, 99.8%, kelong).

## 2.2. Synthesis of perovskite nanocrystals

All the reagents were used directly without further purification. In a typical synthesis of green perovskite nanocrystals, 0.4 mmol CsBr and 0.4 mmol PbBr<sub>2</sub> were dissolved in 10 mL DMF. 1 mL OA and 0.5 mL OAm were loaded into the precursor solution, and the mixture was magnetically stirred. Then, 1 mL of the precursor solution was quickly transferred into 10 mL TEOS under vigorous stirring. Finally, strong green emission was observed immediately after the precursor was injected into TEOS. Other samples with different emission colors were prepared with the other mixture of PbX<sub>2</sub> and CsX (X = I, Br, Cl). Note that all experimental operations in this paper were implemented in the open air. In addition, the ratios of CsBr to PbBr<sub>2</sub> in precursor solution were adjusted as 1:0.33, 1:1, 1:3, 1:5, 1:7 and 1:9, aiming at the transformation of perovskite phase.

# 2.3. Construction of green LED lamp

First, green perovskite powders were dispersed in a small volume of A glue of silicone for 1 h to ensure fine dispersion of the perovskite nanocrystals. Then, the former mixture was added to a large volume of B glue of silicone (A:B = 1:4 wt%), after stirring the mixture for 20 min, nanocrystals/silicone composite was vacuumed to discharge air bubbles. Subsequently, the mixture was casted onto the surface of a UV LED chip (centered at 365 nm), the sample was dried at 60 °C for 40 min to mix the composite thoroughly. Finally, the temperature was increased to 135 °C to cure the nanocrystals/silicone composite and improve the anti-fading effect, and a monochromatic green LED lamp was obtained by the above procedure.

### 2.4. Characterization

Powder x-ray diffraction (XRD) was performed using a Bruker D8 Advance x-ray diffractometer using Cu K $\alpha$  radiation (1.54 Å). Transmission electron microscopy (TEM) study was carried out using a JEOL JEM 2100F field emission transmission electron microscope at 200 kV. Steady state photoluminescence (PL) spectra were acquired using FLS 980 (Edinburgh Instruments).

# 3. Results and discussion

In this work, LTaS approach was utilized to synthesize perovskite material and the schematic illustration of the detailed formation process and mechanism are illustrated in Fig. 1a. Taking bromine-based perovskite as an example, DMF as a good solvent was used to dissolve CsBr and PbBr<sub>2</sub>, and the addition of OA and OAm served to control the size and crystal structure of perovskite nanocrystals. Then, TEOS was exploited as a new antisolvent when halides precursor solution was injected into TEOS, the precursors promptly recrystallized and a fluorescent solution was rapidly observed due to different solubility for halides precursor in two solvents. Generally, TEOS is used as silicon source to form  $SiO_2$  to enhance the stability of perovskites [31,32]. Here, by virtue of the symmetry of molecular structure of TEOS, the low polarity as a solvent can facilitate the formation of a supersaturated condition to form the perovskite nanocrystals. TEM image in Fig. 2b obviously reveal well-proportioned perovskite nanocrystals with the size of around 12 nm. After centrifugation, the as-collected yellow powders displayed pure green emission exposured to UV lamp. As toluene was substituted by TEOS, the room temperature synthetic process is not only green safety but also favorable to scale-up trial. In the latter section of this paper, the optical properties and structural features of the all inorganic perovskite materials prepared by this method will be described in detail.

To validate experimentally the suitability of new antisolvent for perovskite in the visible range, we proceeded with the fabrication of mixed halide perovskite. In Fig. 2a, TEOS as antisolvent, a great quantity of the mixed halide perovskite powders were collected, which emitted a variety of colors extending the entire visible region. XRD measurement was directly used to evaluate the phase composition of the mixed halide perovskite samples. The corresponding XRD patterns of as-synthesized mixed halide perovskite powders are presented in Fig. 2b and compared with patterns from standard Cs<sub>4</sub>PbBr<sub>6</sub> (JCPDS Card No. 73-2478) and CsPbCl<sub>3</sub> (JCPDS Card No. 18-0366). When the ratio of CsBr to PbBr2 was about one, the diffraction peaks were well matched the patterns of standard Cs<sub>4</sub>PbBr<sub>6</sub>. In a wide range of composition, when Br (1.14 Å) atoms were substituted by bigger I atoms (1.33 Å), leading to an evident blue shift of main diffraction peaks. Conversely, interchanged by smaller Cl (0.99 Å) atoms partially, red shift of the diffraction angles was observed. When the ratio of CsCl to PbBr<sub>2</sub> was about one, small shoulder peaks at 21.8°, 31.3° and 38.3° could be indexed to the (101), (200) and (112) planes according to the pattern of standard CsPbCl<sub>3</sub>. This tendency became more obvious as the Cl content continues rising by changing the precursor, CsPbX<sub>3</sub> is the main crystal structure, where the relative intensity of diffractions peaks from CsPbX<sub>3</sub> phase to Cs<sub>4</sub>PbX<sub>6</sub> phase increased when the ratio of CsBr to PbCl<sub>2</sub> was one. XRD analysis indicated that the Cl-rich reaction conditions favor the formation of CsPbX<sub>3</sub> structure, while Br-rich and I-



Fig. 1. Perovskite nanocrystals synthesized by LTaS method. (a) Schematic illustration of the formation process. (b) Typical TEM image of perovskite nanocrystals and photographs of perovskite powder and colloid solution.

rich ones tend to form hexagonal  $Cs_4PbX_6$  phase. This regular variation in components and crystal structure of perovskite is dependent on flexible formation of the diverse products attributed to the applicable choice of antisolvent (TEOS).

Fig. 3 showed the representative TEM and HRTEM images of three typical mixed halide perovskite nanocrystals, prepared with  $CsBr/PbI_2 = 1$ ,  $CsBr/PbBr_2 = 1$ ,  $CsBr/PbCI_2 = 1$ , respectively. In general, the existence of oleylamine and oleic acid as strong binding ligands can control the particle size and shape effectively, and almost

monodispersed perovskite nanocrystals were obtained. HRTEM images showed the bright lattice fringes of individual perovskite nanocrystal, and the well-resolved lattice fringes demonstrate the high crystallinity of the halide perovskite nanocrystals. The lattice spacings were calculated to be 0.283 nm and 0.298 nm for I-Br mixed halide samples, which were consistent with the (006) and (223) lattice planes of  $Cs_4PbX_6$ phase. When the ratio of CsBr to PbCl<sub>2</sub> was one, the calculated lattice space of 0.301 nm corresponded to (200) lattice plane of Br-Cl mixed halide CsPbX<sub>3</sub> nanocrystals in Fig. 3f. These results also evidently



Fig. 2. Structural characterization and photographs of the as-prepared perovskite nanocrystals. (a) Photograph of perovskite powders prepared on a large scale under ambient light and 365 nm UV lamp irradiation. (b) XRD patterns of corresponding perovskite powders.



Fig. 3. TEM and HRTEM images of typical perovskite nanocrystals with different precursors. (a, d)  $CsBr/PbI_2 = 1$ , (b, e)  $CsBr/PbBr_2 = 1$ , and (c, f)  $CsBr/PbCl_2 = 1$ , respectively.

proved that changing halides can achieve high quality and monodispersed nanocrystals with tunable components by the LTaS method.

For the in-depth and comprehensive insight into the luminescent properties of as-prepared perovskite nanocrystals, their emission spectra were investigated and displayed in Fig. 4. By changing the precursors, five typical halide perovskite nanocrystals were obtained which emitted different colors covering visible light range (Fig. 4c). All the samples exhibited a single PL peak, centered at 626 nm, 565 nm, 537 nm, 475 nm and 437 nm, with the emission FWHM of 35 nm, 33 nm, 23 nm, 20 nm and 19 nm, respectively. These narrow FWHM values of PL spectra were definitely beneficial for the fabrication of the monochromatic LEDs with high color purity and high luminous efficiency. In addition, the PL peak demonstrated a remarked blue shift from 626 nm to 437 nm upon the partial replacement of I constituent from Br to Cl, this systematic change caused the variation in the optical gap which made contribution to versatile emission colors. Therefore, as a new and low toxicity antisolvent, TEOS successfully constructed a favorable environment for supersaturated recrystallization to efficiently

guarantee the excellent luminescent performance of all-inorganic perovskite nanocrystals.

In order to determinate of fluorescent phase of the as-prepared all inorganic perovskite nanocrystals, we also probed the influence of the ratios of CsBr to PbBr<sub>2</sub> on the phase transformation of bromine-based perovskite. XRD patterns were used to determine the phases of the asobtained perovskite materials. Fig. 5a showed the XRD patterns of the final products with different ratios of CsBr to PbBr<sub>2</sub> (1:0.33, 1:1, 1:3, 1:5, 1:7 and 1:9). When the molar ratios of CsBr to PbBr<sub>2</sub> are 1:0.33 and 1:1, the XRD patterns were assigned to zero dimensional Cs<sub>4</sub>PbBr<sub>6</sub>. After decreasing the ratio to 1:3, major peaks of the XRD patterns were labeled to three dimensional CsPbBr<sub>3</sub> (JCPDS Card No. 18-0364). When the ratio was further decreased to 1:3 and 1:7, the product was a mixture of CsPb<sub>2</sub>Br<sub>5</sub> (JCPDS Card No. 25-0211) and CsPbBr<sub>3</sub>. With the increased PbBr<sub>2</sub> and reduced CsBr content, the incorporation of PbBr<sub>2</sub> into Cs<sub>4</sub>PbBr<sub>6</sub> crystal structure and dissociation of CsBr from CsPbBr<sub>3</sub> cause phase transformation [30,33,34]. As the ratio continuously dropped to 1:9, an unreported phase has been discovered. Incredibly, it



Fig. 4. Optical properties of the as-prepared perovskite nanocrystals. Photographs of all inorganic perovskite nanocrystals in TEOS under (a) ambient light and (b) 365 nm UV lamp irradiation. (c) Photoluminescence spectra of the five typical samples with different component of halides.



Fig. 5. Structural and luminescent properties of the as-prepared perovskite nanocrystals by controlling the ratios of CsBr to PbBr<sub>2</sub>. (a) XRD patterns of bromine-based perovskite nanocrystals. (b) Photoluminescence spectra of the corresponding perovskite colloidal solution. Inset showing bromine-based perovskite in TEOS under ambient light and 365 nm UV lamp irradiation.

is difficult to determine the phase composition according to the published reference. The characteristic diffraction peaks of this sample do not belong to impurity phases such as CsBr and PbBr<sub>2</sub>, and future work will focus on this phenomenon. As presented in Fig. 5b, the photoluminescence spectra of the corresponding perovskite colloidal solution shifted from 510 nm to 532 nm. Recently, Cs<sub>4</sub>PbBr<sub>6</sub> as a luminescent material is controversial. Mohammed's group has shown that Cs<sub>4</sub>PbBr<sub>6</sub> is highly luminescent with a band gap of 2.33 eV [35,36]. However, a recent work shows that Cs<sub>4</sub>PbBr<sub>6</sub> is a non-luminescent material because of high band gap (3.95 eV), and they think that the reported emissive Cs<sub>4</sub>PbBr<sub>6</sub> is originated from the presence of CsPbBr<sub>3</sub> impurities [30]. Here when the ratio of CsBr to PbBr2 was 1:0.33, the photoluminescence spectra of this sample cannot be measured, which is consistent with the phenomenon shown in the inset of Fig. 5b, confirming that Cs<sub>4</sub>PbBr<sub>6</sub> phase is non-fluorescent. For the sample with the ratio of 1:1, Cs<sub>4</sub>PbBr<sub>6</sub> crystal structure was the major phase and a few CsPbBr<sub>3</sub> nanocrystals may be presented together which emits real fluorescence. In addition, the emission peak of CsPbBr<sub>3</sub> has an obvious blue shift than the Cs<sub>4</sub>PbBr<sub>6</sub> and CsPb<sub>2</sub>Br<sub>5</sub>, suggesting CsPbBr<sub>3</sub> phase has more significant quantum confinement effect than Cs<sub>4</sub>PbBr<sub>6</sub> and CsPb<sub>2</sub>Br<sub>5</sub> phases. In brief, TEOS as an antisolvent can realize the controllable transformation of crystal structure for perovskite.

Finally, in order to explore the practical application of all-inorganic perovskite nanocrystals by using LTaS approach, one monochromatic LED lamp with brightly pure green emission was fabricated as shown in Fig. 6a. This green LED lamp was constructed integrating brominebased nanocrystals with a commercially available 365 nm GaN LED chip, and this LED lamp can work well even after the thermal treatment at 408 K, certainly indicating these all-inorganic perovskite nanocrystals have promising application in the optoelectronics field against harsh environment. Moreover, the narrow emission spectrum strongly confirmed the good monochromaticity of this green LED lamp with the corresponding color coordinate of (0.2931, 0.6928) (Fig. 6c), unambiguously rendering the potentially commercial possibility of allinorganic perovskite nanocrystals green-manufactured by LTaS approach.

# 4. Conclusion

In summary, through the exploitation of TEOS solvent, a low toxicity and suitable antisolvent, we have developed a novel LTaS method to successfully synthetize monodispersed and luminescent all-inorganic perovskite nanocrystals at room temperature. These as-prepared nanocrystals deliver controllably large span emission spectra ranged from 437 nm to 626 nm as well as excellent narrow FWHM (low to 19 nm) monochromaticity. More importantly, only by adjusting the ratios of CsBr to PbBr<sub>2</sub>, this kind of LTaS approach with TEOS can be used to synthesize bromine-based perovskites with different phase structure, Cs<sub>4</sub>PbBr<sub>6</sub>, CsPbBr<sub>3</sub> and CsPb<sub>2</sub>Br<sub>5</sub>. Besides, a bright green LED lamp by a combination of perovskite nanocrystals and UV LED chip has been packaged for the confirmation of its practical application. Hence, TEOS will be a candidate to substitute toluene and hexane for the syntheses of perovskite materials. We believe that this study will pave the way to green-exploit the various all inorganic perovskite materials, and further expand the applications for all-inorganic perovskites, like color converters and LEDs.



Fig. 6. The spectra characteristics of bromine-based perovskite nanocrystals based LED lamp. (a) The picture of green LED lamp. (b) PL emission spectrum and (c) CIE chromaticity coordinate of the green LED lamp.

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## Data statement

The processed data required to reproduce these findings cannot be shared at this time as the data also forms part of an ongoing study.

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